

1 **Complex Modulus of Cold Recycled Mixtures: measurement and modelling**

2 Andrea Graziani¹, Chiara Mignini², Edoardo Bocci³ and Maurizio Bocci⁴

3 (¹ Università Politecnica delle Marche, Via Brece Bianche, Ancona, Italy,
4 a.graziani@univpm.it)

5 (² Università Politecnica delle Marche, Via Brece Bianche, Ancona, Italy,
6 chi.mignini@gmail.com)

7 (³ Università degli Studi eCampus, Via Isimbardi 10, Novedrate (CO), Italy,
8 edoardo.bocci@uniecampus.it)

9 (⁴ Università Politecnica delle Marche, Via Brece Bianche, Ancona, Italy,
10 m.bocci@univpm.it)

11 **ABSTRACT**

12 In the present study, the complex modulus E^* of three types of cold recycled mixtures was
13 measured in the laboratory using cyclic compression tests. The mixtures, that were produced in
14 situ using the full-depth reclamation technique, were characterised by the same 100% reclaimed
15 aggregate composition, whereas bitumen emulsion, foamed bitumen and portland cement were
16 employed as binding agents. Complex modulus testing was carried out on cylindrical specimens
17 cored from the pavement seven years after construction. A sinusoidal strain with amplitude of
18 30 microstrain was applied at testing temperatures ranging from 0 to 50 °C and frequencies ranging
19 from 0.1 to 20 Hz. Results showed that the time-temperature superposition principle was fully
20 verified for the stiffness modulus whereas some discrepancy was highlighted by the loss angle.
21 The Huet-Sayegh rheological model was applied to simulate the experimental data. Model fitting
22 based only on the stiffness modulus provided excellent results. Indeed, the fitted rheological model
23 was also able to simulate the loss angle variation quite well, but with some underestimation (about
24 3°) of the measured values. It can be concluded that other dissipation mechanism, in addition to
25 the “classical” viscous type, were influent for behaviour of the studied cold recycled mixtures.

26
27 **Keywords:** Cold recycling, bitumen emulsion, foamed bitumen, complex modulus, Huet-
28 Sayegh model.

30 **1. INTRODUCTION**

31 Cold recycling technologies have been developed over the last 20 years as a cost-effective
32 and sustainable alternative for the rehabilitation of asphalt pavements. Among these technologies,
33 the full-depth reclamation (FDR) allows 100% recycling of the existing pavement materials [1,2].
34 The aggregate blend of cold recycled mixtures (CRM) produced in-place using FDR is composed
35 mainly of reclaimed asphalt pavement (RAP) and reclaimed aggregate, the latter coming from the
36 unbound or cement bound layers. The other CRM components are virgin aggregate (added to
37 adjust the grading curve), water and different types of binders [3].

38 When only cement is used, the CRM is actually a cement-treated material (CTM) [4-5].
39 Whereas, if both cement and bitumen (in form of emulsion or foam) are used a bitumen-stabilized
40 material (BSM) or a cement-bitumen treated material (CBTM) is obtained (for BSM the

1 bitumen/cement ratio is greater than one and cement content is lower than 1% while for CBTM
2 the bitumen/cement ratio is greater than one and cement content is greater than 1%) [6-8].

3 The mechanical properties of CRM depend on the nature and proportion of the binders.
4 They may range from a stiff and brittle behaviour (typical of CTM), to a stress-dependent granular-
5 like behaviour (typical of BSM), to an asphalt-like behaviour with cumulative damage failure
6 (typical of CBTM) [9]. Besides their mechanical response, it is underlined that, because of the
7 presence of water and cement, CRM are subjected to a curing process, which entails an evolution
8 of their physical and mechanical properties over time [10].

9 In recent years, many researchers investigated CRM properties in terms of stiffness,
10 resistance and fatigue behaviour [11-15]. In particular, various studies focused on the evaluation
11 of the rheological properties of CRM, which are strictly related to the presence of bitumen in the
12 mix [8, 16-18]. Indeed, the CRM inherit the thermo-dependency and the viscoelastic behaviour of
13 the bitumen, both from the fresh binding agent (bitumen emulsion or foamed bitumen) and from
14 the RAP. In particular, it has been shown [16] that the aged and oxidized bitumen in the RAP
15 determines a limited, but yet measurable, time- and temperature-dependent behaviour in cold
16 recycled CTM.

17 The present paper focuses the linear viscoelastic (LVE) characterization of different CRM,
18 produced in-place with the FDR technology, through the measurement and modelling of the
19 complex Young's modulus E^* . The same experimental approach and analytical models normally
20 used for hot mix asphalt (HMA) are applied to CRM produced using different binders: only
21 cement, bitumen emulsion and cement, foamed bitumen and cement.
22

23 **2. EXPERIMENTAL PROGRAM**

24 **2.1 Materials**

25 The cores tested in this study were extracted from an experimental pavement section
26 constructed in 2007 during the rehabilitation of the SS38 highway in the north of Italy. In the
27 project, the existing HMA layers were milled and the RAP was used to produce a new 300 mm
28 CRM layer above the subgrade. The experimental pavement was divided into three subsections,
29 where the reclaimed aggregates were stabilised with the FDR technology using different binders:
30 foamed bitumen and cement (CRM-FB), bitumen emulsion and cement (CRM-EM) and only
31 cement (CRM-PC). Finally, three HMA courses (total thickness of 190 mm) were placed on top
32 of the CRM layer.

33 The foamed bitumen was produced with a recycler machine model WR 2000, combining
34 a 50/70 pen bitumen with 3% of foamant water (by bitumen weight). The bitumen emulsion was
35 an over-stabilised cationic type, specifically designed for cold in-place recycling, with designation
36 C 60 B 10, according to EN 13808. The cement was a pozzolan-lime type IV/A, strength class
37 32.5 R (EN 197-1). The binder dosages (by dry aggregate weight) adopted in the three subsections
38 are reported in Table 1.

39 Seven years after construction, 100 mm diameter full-depth cores were taken from the
40 pavement. The CRM and HMA layers were sawed in order to obtain cylindrical specimens for
41 complex modulus testing. The air voids characteristics of the specimens are shown in Table 1.
42

2.2 Complex modulus testing

The complex modulus was measured using a servo-hydraulic testing system AMPT PRO. Axial stress was measured with a load cell, whereas axial strain was measured on the middle part of the specimen using three LVDT, placed 120° apart, with a measuring base of 70 mm. A haversine compression loading was applied in order to obtain an axial strain with a target amplitude of 30 microstrain. Testing temperatures ranged from 0 to 50 °C and frequencies ranged from 0.1 to 20 Hz; 20 loading cycles were applied at each frequency.

The steady-state sinusoidal component of stress and strain were employed to calculate the complex Young's modulus:

$$E^* = \frac{\sigma_0 \exp[j(\omega\tau + \phi)]}{\epsilon_0 \exp[j(\omega\tau)]} = \frac{\sigma_0}{\epsilon_0} \exp[j\phi] = E_0 \exp[j\phi] = E_1 + jE_2 \quad (1)$$

where j is the imaginary unit, σ_0 and ϵ_0 are the steady state amplitudes of stress and strain, respectively, E_0 is the absolute value of E^* (also indicated as stiffness modulus), ϕ is its loss (or phase) angle, and E_1 , E_2 are its storage and loss components, respectively.

TABLE 1 Properties of the tested cores

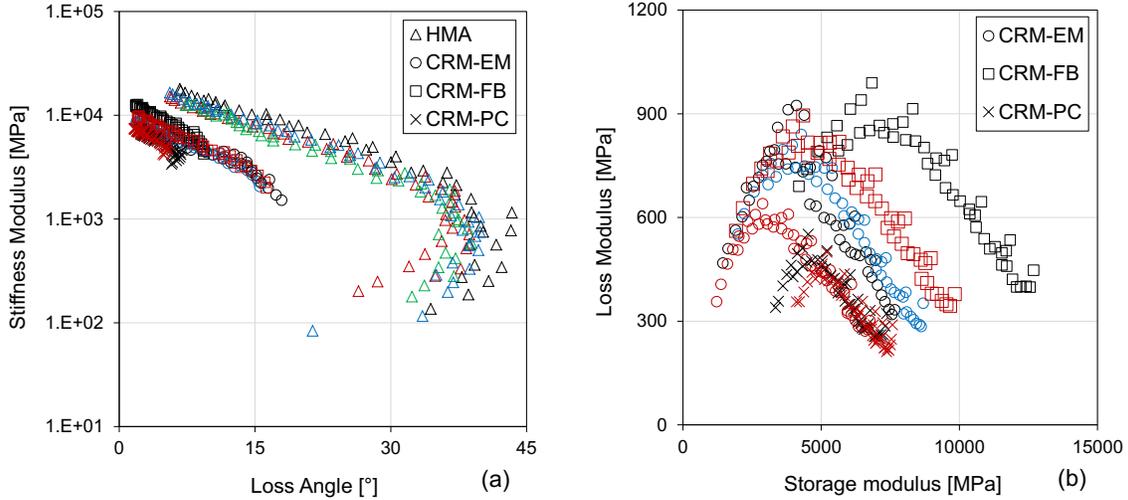
Mixture	Bitumen content [%]	Cement content [%]	Specimen ID	Air voids [%]
Hot Mix Asphalt	4.50 (Hot binder)	/	HMA1	7.0
			HMA2	9.0
			HMA3	7.5
			HMA4	7.5
CRM-Emulsion	1.80 (Emulsion)	2.00	CRM-EM1	17.5
			CRM-EM2	18.9
			CRM-EM3	18.9
CRM-Foam	3.00 (Foam)	1.75	CRM-FB1	9.6
			CRM-FB2	14.1
CRM-Cement	/	3.00	CRM-PC1	13.7
			CRM-PC2	14.7

3. RESULTS AND ANALYSIS

The values of E_0 and ϕ measured on all tested specimens are shown in the Black diagram (Figure 1a). In this plot it can be noticed that E_0 varies from 17931 to 84 MPa and from 12694 to 1521 MPa, for HMA and CRM, respectively. Also, ϕ ranges from 5.6 to 43° and from 1.8 to 18°, for HMA and CRM, respectively. Although the results show that the complex modulus of all CRM specimens is clearly thermo- and frequency-dependent, the variability of both E_0 and ϕ values throughout the tested temperature and frequency ranges is limited, with respect to the variability which is typical of HMA specimens.

In order to highlight the differences in the LVE behaviour among the different CRM, the values of E_1 and E_2 are plotted in the Cole-Cole diagram (Figure 1b). In this plot it can be noticed that, as expected, CRM-EM and CRM-FB specimens showed a higher variability with temperature and frequency, with respect to CRM-PC.

1 It is remarked that such behaviour does not imply any form of blending between aged
 2 bitumen from RAP and fresh bitumen (emulsion residue or foamed bitumen). In particular, when
 3 bitumen emulsion is used, the bitumen droplets coalesce forming a continuous film that adheres to
 4 larger RAP particles or includes smaller RAR particles. However, the two bituminous phases are
 5 always at ambient temperature, which makes their physical blending quite difficult.
 6



7 **FIGURE 1 The measured values of E^* : a) Black diagram; b) Cole-Cole diagram**

8
 9
 10 It is well known that, if the LVE behaviour of a material can be represented by a unique
 11 curve in the Black and Cole-Cole diagrams, validity of the time-temperature superposition
 12 principle (TTSP) is confirmed and the behaviour can be considered thermo-rheologically simple.

13 For HMA specimens, some dispersion is present in the measured E^* values, however the
 14 experiments confirm the validity of the TTSP (the small deviations may be due to the effect of
 15 aging). On the other hand, the behaviour of CRM should be analysed from a different perspective.
 16 In fact, for these mixtures, the simultaneous presence of aged binder (RAP), fresh binders (bitumen
 17 emulsion residue and foamed bitumen) and cement, makes the a-priori assumption of thermo-
 18 rheological simplicity difficult.

19 Notwithstanding the aforementioned issues, master curves of the stiffness modulus E_0 at
 20 the reference temperature $T_0 = 0$ °C were obtained for all tested specimens (Figure 2). In particular,
 21 the temperature shift factors were determined according to the closed form shifting (CFS)
 22 algorithm [19]. The CFS algorithm is based on the definition of an overlapping window between
 23 two successive isothermal curves: when the area between the two curves is zero the optimal
 24 shifting is obtained.

25 As it can be observed in Figure 2, validity the TTSP for E_0 is perfectly confirmed for all
 26 tested specimens. This suggests that the same rheological model could be used to simulate the
 27 LVE behaviour of HMA and CRM. The Huet-Sayegh (HS) model [20] was employed:
 28

$$E^* = E_e + \frac{E_g - E_e}{1 + \delta(j\omega\tau)^{-k} + (j\omega\tau)^{-h}} \quad (2)$$

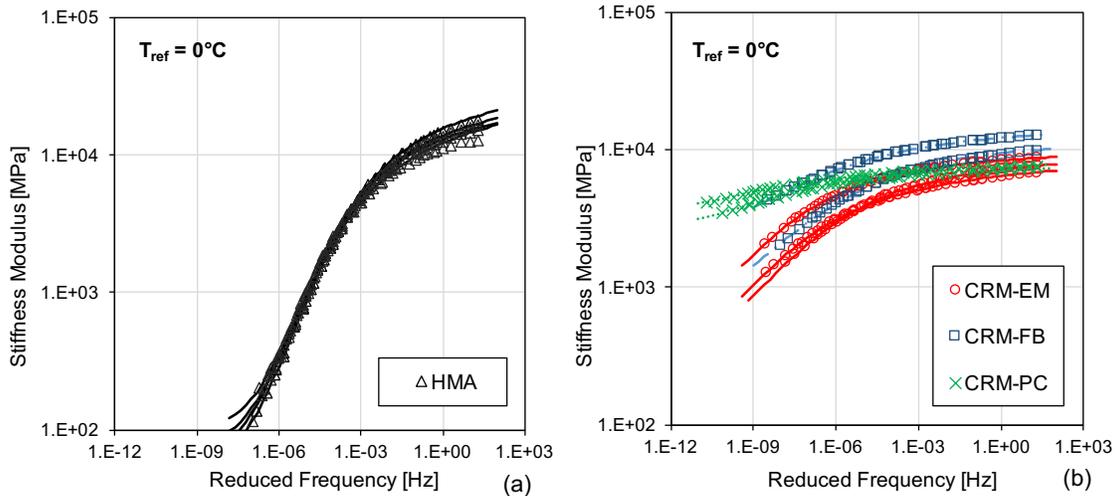
29

1 where E_g , E_e are the glass and equilibrium moduli, respectively, δ , h and k are dimensionless
 2 model parameters ($k < h < 1$), $\omega = 2\pi f$ is the angular frequency (f is the testing frequency in Hz)
 3 and τ is the characteristic time. In particular, τ is a function of the testing temperature T :

$$\tau(T) = a(T)\tau_0 \quad (3)$$

4
 5
 6 where $a(T)$ are the temperature shift factors calculated using the CFS algorithm and τ_0 is the
 7 characteristic time at the reference temperature [19]. Lines representing the fitted HS model are
 8 superposed to the measured E_0 values in Figure 2. Table 2 lists the HS model parameter, and the
 9 parameters (C_1 , C_2) that were used to fit the shift factors with the Williams-Landel-Ferry model
 10 [21].

11 As it can be observed in Figure 2a, the behaviour of all HMA specimens can be simulated
 12 using identical values of h and k ; the values of δ and $\log \tau_0$ are also very close. It can also be
 13 observed that E_g values are correlated to the air voids content of the specimens reported in Table
 14 1 (higher voids correspond to lower E_g).



16 **FIGURE 2 Master curves of the stiffness modulus at 0°C: a) HMA; b) CRM**

17 **TABLE 2 Huet-Sayegh and WLF model parameters ($T_{ref} = 0^\circ\text{C}$)**

Specimen ID	E_g [MPa]	E_e [MPa]	k [-]	h [-]	δ [-]	$\log \tau_0$ [-]	C_1 [-]	C_2 [°C]
HMA1	30000	50			2.13	1.49		
HMA2	23000	100	0.16	0.55	2.00	1.80	19.9	116.1
HMA3	24000	75			2.48	1.76		
HMA4	26000	60			2.22	1.72		
CRM-EM1	9500	200			1.02	4.77		
CRM-EM2	8500	80	0.09	0.30	1.22	5.65	23.1	109.7
CRM-EM3	10000	150			0.86	6.16		
CRM-FB1	15000	1000			0.78	6.05		
CRM-FB2	11300	20	0.08	0.22	0.35	4.46	32.3	174.1
CRM-PC1	8200	2500			0.89	7.43		
CRM-PC1	8800	3200	0.07	0.29	1.26	8.17	25.7	87.0

Figure 2b shows that the HS model also simulates correctly E_0 measured on CRM specimens. Similar to HMA, unique values of the parameters h and k were estimated for each CRM and both parameters are significantly lower with respect to HMA. As expected, the CRM-PC specimens are characterised by the highest values of E_e . The estimated values of E_g are quite dispersed, but this is not surprising since the CRM were produce on site using FDR and dispersion is also present in their volumetric properties (Table 1). It is worth noting that τ_0 gives a consistent ranking of the relaxation ability of the different materials. Specifically, τ_0 of CRM-EM and CRM-PC is about 4 to 5 orders of magnitude higher with respect to HMA, and about two order of magnitude lower than CRM-PC.

Figure 3 reports the master curves of the loss angle, along with the fitted HS models. The model simulates well the pattern described by the experimental data and the observed differences are generally less than 3° . In Figure 3b, it can also be noted that, for all CRM, the HS model systematically underestimates the loss angle. This suggests that in these materials a non-negligible part of energy dissipation is due to physical mechanisms which are not captured by the HS model. Possible explanations could be non-linearity or frictional dissipation due to aggregate-to-aggregate contact (in CRM, the aggregate particles are not completely covered by bitumen). However, additional testing is necessary to investigate the dissipation mechanisms taking place inside CRM.

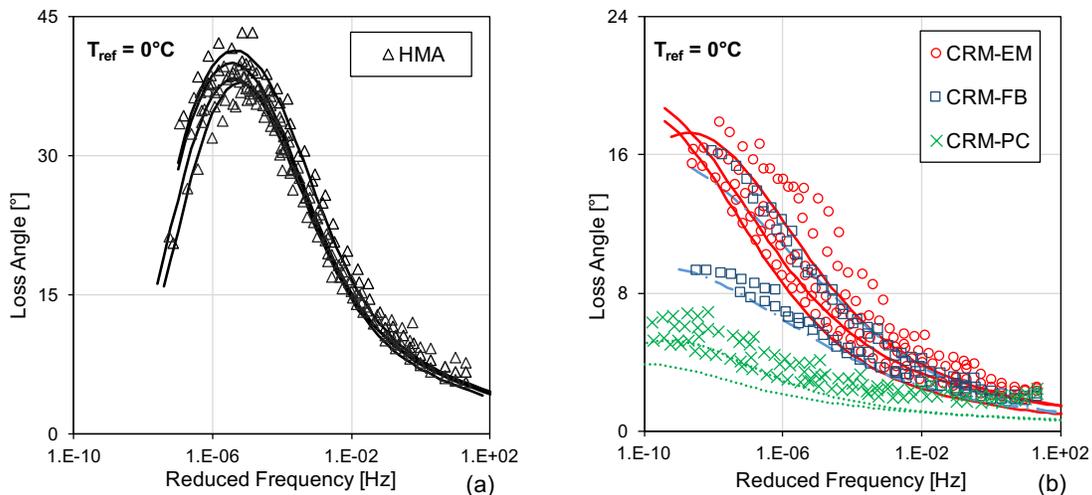


FIGURE 3 Master curves of the loss angle at 0°C : a) HMA; b) CRM

4. CONCLUSIONS

Based on the experimental results and analysis it is possible to conclude that the time-temperature superposition principle can be applied to analyse the LVE behaviour of CRM produced with bitumen emulsion and cement, foamed bitumen and cement and only with cement.

As regards the application of the Huet-Sayegh model it is possible to conclude that:

- The model can be used to simulate the LVE behaviour of CRM, obtaining an excellent fitting, in particular with the stiffness modulus data;
- The parameters of the model can be related to the physical and volumetric properties of both HMA and CRM specimens;
- The model underestimates the loss angle of CRM, probably because it can only capture linear and viscous dissipation mechanisms.

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